

Anisotropy of Fractal Dimensions of Fractures and Loading Curves of Steel Samples During Impact Bending

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Abstract: This study aims to establish the correlation between the impact strength and texture, fractal dimensions of fractures (D_f), fractal dimensions (D_c) obtained from load-time diagrams $P(\tau)$ reflecting the applied load (P) dependence on time (τ) during the Charpy impact test of 20K steel at various temperatures as well as the comparison of the abovementioned fractal dimensions. The tests were carried out on a vertical impact testing machine with a multi-channel system for high-speed registration of forces and strains, as well as a heating and cooling system for samples in a wide temperature range. The load vs. time (load dependence on time) diagrams were obtained at an impact velocity of $V_0 = 4.4$ m/s at temperatures of -50, +20, +50 °C. The Charpy standard samples of 20K steel (analog to DIN17175, class St45.8) were cut in various directions out of a 12 mm thick the destroyed tank shell of a distillation column for oil refining. It was established that the behavior of both the abovementioned fractal dimensions depending on the cutting direction and test temperature coincides qualitatively. The trend of decreasing in fractal dimension with a more viscous nature of fracture was found. The effect of texture is discussed.

Keywords: impact tests, diagram of load changes depending on time, impact toughness, fractal dimension, texture.

1. INTRODUCTION

The Charpy impact tests allow finding the characteristics which are important for structural materials, such as impact strength, brittle-ductile transition temperature, that is, the temperature at which the fracture of the material during loading changes from viscous to brittle. The above data are essential for predicting the safe operation of the designed structures. The ductile-brittle transition temperature (DBTT) using Charpy instrumental tests can be determined by various methods. Among them, there can be, for example, the lateral expansion, the appearance of shear fracture, the average value of impact strength, the use of a load diagram, a master curve [1]. To assess the resistance to brittle fracture of structural materials, in particular steel, impact tests (KCV) of a series of Charpy samples are usually carried out at various temperatures. At this method, the study of fracture surfaces is a very important process, because it allows you to get an additional estimate of the temperature of the plastic-brittle transition. The critical temperature for the transition of a material from a viscous state to a

brittle one (T_{br}) is determined from the graphical impact strength dependence on temperature. One of the methods for determining the critical temperature of brittleness is based on the fact that the proportion of brittle and viscous component is in the ratio "50:50" at this temperature [2-4].

A significant scatter of the obtained data takes place at using various equipment for impact testing and different assessment methods [5-7].

The texture of polycrystalline bodies is the main cause of the anisotropy of their physical and mechanical properties. Therefore, the presence of texture in the material is also one of the reasons for this scattering [8-10]. Regarding this, it is also necessary to take into account the cutting direction of samples from sheets or tubular fragments of the study materials for physical research and mechanical testing.

The use of modern techniques and high-speed systems for recording diagrams, which reflect the applied load dependence on the fracture time, makes it possible to increase the informational content of impact strength tests [11-13].

Additional information can also be obtained by studying the relationship between the impact

toughness and fractal dimension (FD) of the fractures [14, 15].

It should be noted that the description of fracture surfaces in the language of fractals and the correlation connection with the FD index and the mechanical properties of materials have still been being debated [16].

It should be noted that the description of fracture surfaces in the language of fractals and the correlation connection the FD index and the mechanical properties of materials have still been being debated [16]. So, for example, in [17, 18], the correlation between the crack fractal dimension, the value of the stress intensity factor, and the structure of the pre-fracture zone with an invariant complex of mechanical properties are indicated. The correlation connection with the fractal dimension of fractures and the durability of the tested samples is traced in [14]. As durability decreases, the value of the fractal dimension increases. It is shown that the FD increases with increasing the number of cycles to failure. In [19] and several other earlier studies, an extensive review of which is also presented in [19], it was shown that ductile alloys show a decrease in fractal dimension with increasing viscosity, while brittle alloys demonstrate the opposite behavior.

However, not all studies using impact toughness are concordant. In [20], it is questioned that a fracture is a self-similar object in the case of viscous fracture.

This study aims to establish the correlation (correlative values) between the impact toughness and the texture, fractal dimension (D_f) of fractures, fractal dimensions (D_c) determined from load-time diagrams $P(\tau)$ reflecting the applied load (P) dependence on time (τ) during the Charpy impact test of 20K steel at various temperatures as well as the comparison of the abovementioned fractal dimensions.

2. EXPERIMENTAL PROCEDURE

The material for the study was a fragment of steel 20K (analog to DIN17175, class St45.8), cut out of the destroyed tank shell of a distillation column for oil refining. The diameter of the tank shell was 2200 mm including wall thickness of 12 mm. The chemical composition of steel: 0.19 % C; 0.27 % Si; 0.65 % Mn; 0.2 % Cu; 0.23 % Ni; 0.23% Cr; 0.25 % As; 0.05 % S;

0.035% P; Fe is balance in mass percent. Three groups of the Charpy standard samples cut in different directions out of the abovementioned steel fragment were used to study the anisotropy of the impact toughness. The first group of samples was cut along the longitudinal direction (LD), the second group was cut perpendicular to the longitudinal direction, i.e. in the transverse direction (TD), and the third group - at an angle of 45° to RD, i.e. in the diagonal direction (DD), as shown in Fig. 1. The Charpy standard samples with a V-shaped notch had dimension (55×10×10) mm.

Impact tests were carried out¹ by an impact tension machine, equipped with a multichannel system for recording forces and strains, as well as a system for heating and cooling samples in a wide temperature range [21, 22]. The impact velocity V_0 was 4.4 m/s.

The high sensitivity of the recording system allows us to divide the load diagram into characteristic areas and calculate the values of the total deformation and fracture energy and its components: crack initiation energy, viscous crack growth energy, brittle crack penetration energy, and viscous crushing energy [22].



Fig. 1. Scheme of specimens cutting out of the tubular fragment of the steel 20K by length of 2200 mm and thickness of 12 mm for impact tests

The specific fracture energy W_{sp} (KCV) was determined by the formula [23]:

$$W_{sp} = W/S_0 \quad (1)$$

¹ Tests were carried out by Kondryakov E.A.

Where W is the energy of destruction, S_0 is the original cross-sectional area of the sample in the place of the stress concentrator, m^2 , which calculated by the formula [23]

$$S_0 = H_1 B \quad (2)$$

Where H_1 is the original height of the sample's working part, m;

B is the original width of the sample, m. The B and H_1 values are measured with an error that is not more than ± 0.05 mm [23].

The JSM-840 scanning electron microscope in the secondary (SEI) electron mode at an accelerating voltage of 20 kV and magnifications $\times 10 \dots \times 1000$ was used to study the morphology of fractures.

X-ray method by means of inverses pole figures (IPFs) was used for studying the texture. We carried out the θ -2 θ scanning by the DRON-3 m diffractometer by the Bragg-Brentano geometry of investigated samples in the plane perpendicular to the normal direction (ND) to the steel surface and in the plane perpendicular to the longitudinal direction (LD). The sample without the texture (standard) was also scanned under identical geometric conditions. The standard sample was made of fine sawdust after its recrystallization during the vacuum annealing for 1 hour at 450 °C. The Morris normalization was used at the IPFs construction [24].

The fractal dimension D was determined by the box method. An illustration of this method is shown in Fig. 2.



Fig. 2. Illustration of using the method of cells (box method) to the curve fractal dimension determination

The essence of this method is that to determine the fractal dimension it's necessary to cover the image of a curved profile or a selected fracture fragment border with elementary square grids with sides of l_i [25]. At each stage of this method application, the same curve is covered with cells of a reduced scale. The smaller the cell size, the more accurately the curve is reproduced. At the same time, we count the number of squares $N(l_i)$ that the curve intersects. Then we change the size of the grid window l_i . Again we count the number of squares that were crossed by the curve: $N(l_2)$, $N(l_3)$, ..., $N(l_n)$. The number of squares $N(l_i)$, which were crossed by the curve is connected with the size of the grid window l_i by the formula [25]:

$$N(l) = \lim_{l \rightarrow 0} \alpha \cdot l^{-D} \quad (3)$$

Where $D = \lim_{l \rightarrow 0} \frac{\ln N(l)}{\ln(1/l)}$ by definition, is usually called the fractal dimension or the Hausdorff-Besikovich dimension [25].

Practically D is found by the tangent of angle slope of the graphical dependence $\lg N(l_i) = f(\lg l_i)$ [25].

To determine the fractal dimension, HarFA (Harmonic and Fractal Image Analysis) computer software was used [26] (Fig. 3).

Before determining the fractal dimension, the fracture photographs were cleaned from noise using the ACDSee Photo Studio Software [27], to obtain only the boundary lines of the fracture fragments (Fig. 3).

3. RESULTS AND DISCUSSION

In the studied fragment of 20K steel takes place the impact toughness anisotropy (Table 1). Samples cut out in the longitudinal direction (LD), have the least resistance to destruction. Samples cut out in the transverse direction (TD) have the maximum value of the impact toughness. Samples, cut out at an angle of 45° to the LD (i.e. in the DD) have the intermediate value of the impact toughness but the difference between the two latter is minor.

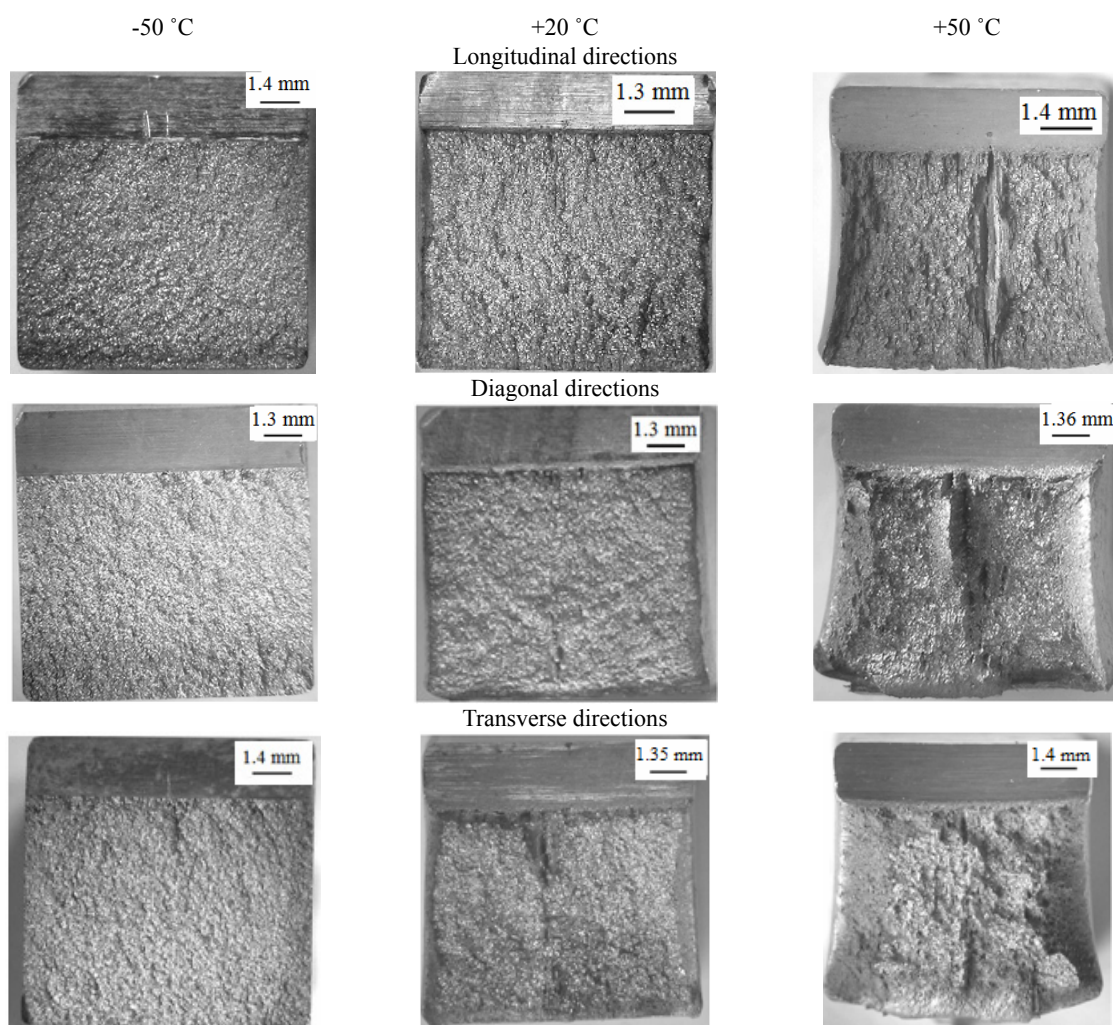


Fig. 3. An example of using the Harfa computer program.

Table 1. The specific fracture energy W_{sp} , J/m², at the impact tests.

Specimen, №	-50°C	+20°C	+50°C
Longitudinal Direction LD			
1	0.83	5.04	22.87
2	0.73	4.31	20.82
Average value	0.78	4.67	21.84
Transverse direction TD			
1	0.97	9.88	47.95
2	0.95	10.91	57.95
Average value	0.96	10.39	52.95
Diagonal direction DD (LD+45°)			
1	0.85	13.33	47.24
2	0.91	17.70	54.65
Average value	0.88	15.51	50.94

Micrographs of fractures are shown in Figures 4, 5. Preliminary analysis of presented micrograms was carried out with a slight increase to determine the general nature of the destruction (Fig. 4). This made it possible to accurately determine the position of the areas of brittle and viscous destruction, as well as the beginning of destruction. As is well known, the energy of the nucleation of destruction, all other conditions being the same, is determined by the temperature and the microstructure nature. The destruction under the notch basically begins as viscous even at negative temperatures due to the peculiarities of the complex-stressed state at the top of the notch (Fig. 4). As can be seen, fractures in this zone have ductile elements of the destruction (bridges and individual viscous pits) as well as deep secondary cracks running from the notch and crossing the entire surface of the fracture, down to the rupture (Fig. 4).

**Fig. 4.** SEM photography of fractures surfaces after impact tests at various temperatures of 20K steel samples cut in the longitudinal, diagonal and transverse directions.

Secondary cracks are most pronounced in the sample cut out in the LD (Fig. 4, c, f). The formation of such cracks in the fracture indicates the presence in the metal of the texture, which determines the resistance to viscous fracture (Table 1).

The metal is destroyed by the mechanism of quasi-cleavage in the whole range of tests temperatures in all directions as further studies have shown (Fig. 5).

Facets of quasi-cleavage have sizes 10 ... 40 microns. The fraction of the viscous component in quasi-cleavage increases with an increase of the test temperature.

Fig. 6 shows the diagrams dependencies of the applied force P on the time τ in the process of impact tests at different temperatures of

specimens cut out in various directions from the fragment of 20K steel.

Magnitudes of fractures fractal dimensions D_f of studied samples, as well as the fractal dimensions of diagrams, $P(\tau)$ are presented in tables 2, 3. Fractal dimensions of both D_f and D_c have minimal values at elevated temperature of tests + 50 °C (tables 2, 3).

At the same time, the impact toughness at test temperature + 50 °C is maximal (Table 1). This indicates the viscous nature of the destruction at + 50 °C. Maximal values of D_f and D_c take place at a low test temperature of -50 °C (tables 2, 3). At the same temperature, the minimal value of impact toughness takes place (table 1), which corresponds to the brittle character of the destruction of study samples at the -50 °C.



Fig. 5. SEM photographs of fractures surfaces after impact tests at various temperatures of 20K steel samples cut in the longitudinal, diagonal and transverse directions.

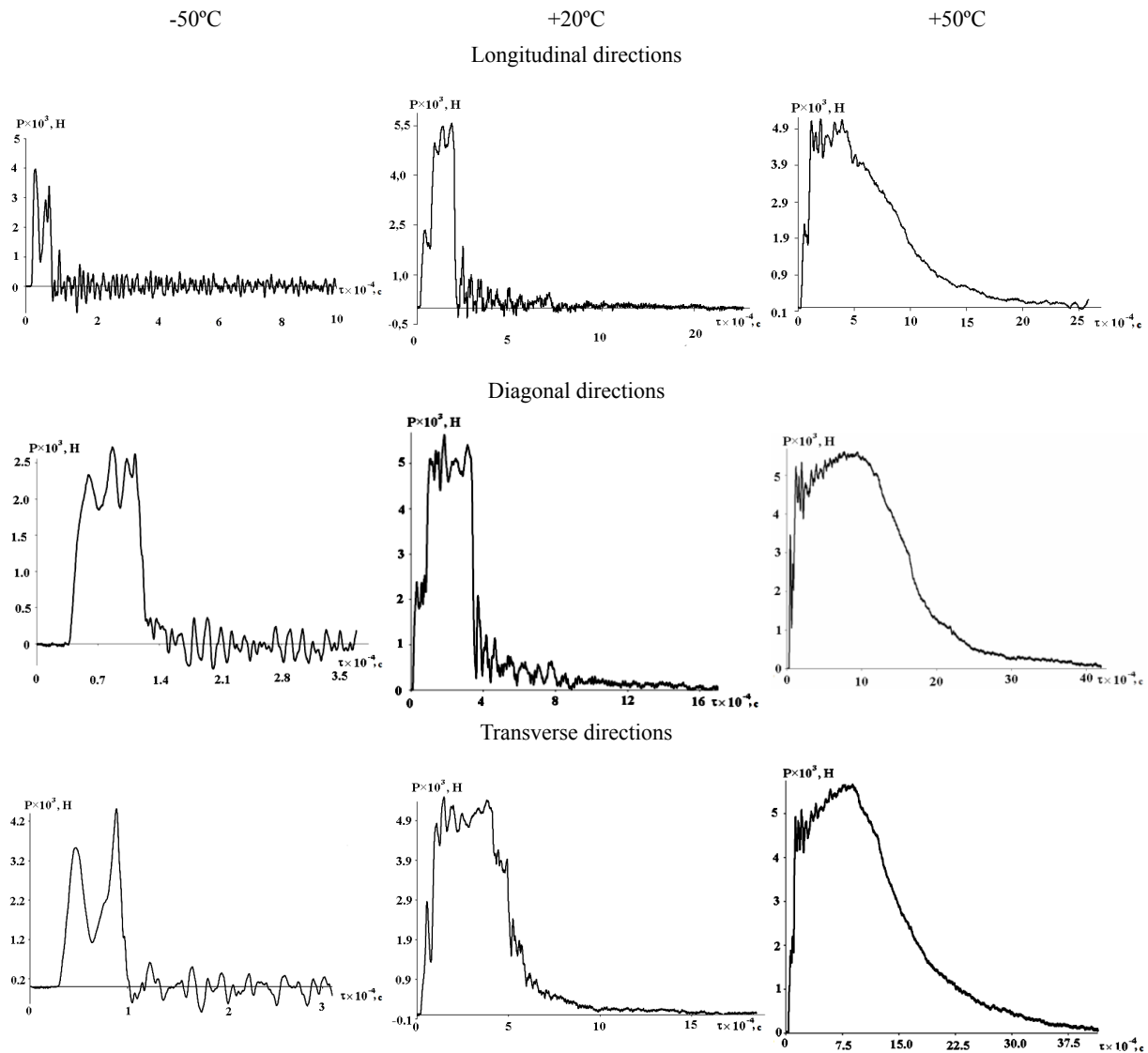


Fig. 6. Diagrams of the applied load (P) from time (τ) during impact test at various temperatures of 20K steel samples cut in the longitudinal, diagonal and transverse directions.

Table 2. The fractal dimension (D_f) of the fractures shown in Figure 5

$t, ^\circ\text{C}$	LD	TD	DD	Average D_f
-50	1.59 ± 0.01	1.60 ± 0.01	1.64 ± 0.01	1.61 ± 0.01
+20	1.55 ± 0.01	1.56 ± 0.01	1.59 ± 0.01	1.57 ± 0.01
+50	1.52 ± 0.01	1.54 ± 0.01	1.55 ± 0.01	1.54 ± 0.01

Table 3. The fractal dimension (D_c) of the $P(\tau)$ diagrams shown in Figure 7

$T, ^\circ\text{C}$	RD	TD	DD	Average D_c
-50	1.27 ± 0.01	1.23 ± 0.01	1.31 ± 0.01	1.27 ± 0.01
+20	1.23 ± 0.01	1.17 ± 0.01	1.26 ± 0.01	1.22 ± 0.01
+50	1.12 ± 0.01	1.13 ± 0.01	1.16 ± 0.01	1.14 ± 0.01

With increasing fracture toughness (table 1), the fractal dimension of the fracture surface decreases (tables 2 and 3). Similar results were

obtained in [19]. In particular, at testing a series of 24 samples of AISI 4340 steel, it was shown that the fractal dimension of the fracture surface

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